

NON-SQUARE QUANTUM WELL GROWTH FOR REDUCED THRESHOLD CURRENT IN TENSILELY STRAINED LASERS OPERATING AT $1.52\mu\text{m}$

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ABSTRACT: *This paper presents calculations demonstrating that non-square quantum well growth (well shaping) can result in reduced threshold current for tensilely strained quantum well bipolar diode lasers operating at $1.52\mu\text{m}$. Calculations of subband structure, optical matrix elements and laser gain are performed for arbitrarily shaped quantum wells based on a 4-band (electron/heavy-hole/light-hole/split off-hole) Hamiltonian. For long wavelength ($1.3\mu\text{m}$ to $1.55\mu\text{m}$) lasers, Auger recombination dominates the threshold current. Compared to a $1.52\mu\text{m}$ optimal square well just below critical thickness, an InGaAs-InGaAsP (on InP) well incorporating potential 'spikes' and having the same wavelength can be much wider. The wider well, possible with well shaping, results in a lower value for three-dimensional (3D) carrier density at a given value of modal gain. For low loss lasers, this implies a reduction in Auger (and hence total) threshold current to a value below the best obtainable in a laser based on a square quantum well.*

INTRODUCTION

Long wavelength ($1.3\mu\text{m}$ to $1.55\mu\text{m}$) strained quantum well (QW) laser diodes have been widely studied since they are key components for optical communication systems. Incorporating strain into the active region of the laser gives such benefits as reduced threshold current, high modulation speed, smaller linewidth enhancement factor and reduced temperature sensitivity. The material of choice for lasers operating at $\sim 1.55\mu\text{m}$ is InGaAs-InGaAsP (on InP) for which the benefits of strain has been proven. It has been reported that tensilely strained quantum wells perform as well as if not better than compressively strained quantum wells [1]. Benefits derived from use of tensile strain have been attributed to reduced level of spontaneous emission and an enhancement of the transition matrix [2, 3]. On the other hand, the technique of non-square quantum well growth (well shaping) has mainly been applied to the enhancement of photo-detectors [4], optical modulators [5, 6], inter-subband (unipolar) lasers [7] and non-linear devices [8, 9]. In recent calculations [10], it was shown that combining strain with an appropriately shaped compressively strained stepped alloy quantum well can lead to reduced threshold current in bipolar laser diodes. This work extends the technique of well shaping to include lasers incorporating tensilely strained quantum wells.

Tensilely strained InGaAs-InGaAsP square quantum well (SQW) lasers having a fixed emission wavelength show a reduction in threshold current as the well width is increased within the limits of critical layer thickness [1]. Silver and O'Reilly [11] have suggested that the increased well width plays a part in reducing the three-dimensional (3D) carrier density in the active region of these lasers, thereby reducing the Auger recombination rate. The minimum threshold current is achieved just before the well width reaches critical thickness. It has now been firmly established that Auger recombination is responsible for up to 80% of the threshold current in long-wavelength (1.3 to $1.55\mu\text{m}$) InGaAs-InGaAsP lasers operating at room temperature [12, 13, 14].

Incorporation of one or two narrow potential spikes into a square quantum well while at the same time keeping the well width fixed results in an increase in the fundamental electron-hole transition energy. On the other hand, increasing the well width of such a 'spiked' quantum well has the counter effect of reducing the transition energy. For a fixed transition energy therefore, a spiked quantum well will be much wider than a conventional square quantum well. In this paper, it is shown computationally that the wider spiked quantum wells have reduced three-dimensional (3D) carrier density at threshold when used in the active region of bipolar diode lasers. This leads to a reduction in the dominant Auger current component.

The paper is organised as follows. First the theoretical model used in the calculation of the quantum well energy bandstructures, optical gain and radiative current density is described. The results are then presented and discussed, followed by the conclusions of the study.

2. THEORETICAL MODEL

Band structure Calculations for a Strained Quantum Well

Consider a quantum well of width L_z grown along z with a biaxial strain. For a direct bandgap semiconductor, the bandstructure near the zone centre can be described by the $\mathbf{k}\cdot\mathbf{p}$ method [15]. In this work, an 8-band effective mass Hamiltonian based on the $\mathbf{k}\times\mathbf{p}$ method is used to describe the coupled QW electron (EL), heavy-hole (HH), light-hole (LH) and split-off-hole (SO) states. By use of a unitary transformation, this 8x8 Hamiltonian is block diagonalized (decoupled) into two 4x4 Hamiltonians whose wavefunctions are given by [16]

$$\Psi_{nk_{\parallel}}^{\sigma}(\mathbf{r}) = \frac{1}{\sqrt{A}} \sum_v F_n^{(v)}(k_{\parallel}, z) e^{ik_{\parallel}\cdot\vec{\rho}} |u_v'\rangle \quad (1)$$

where $s = +ve$ for the upper-block and $s = -ve$ for the lower-block. The summations are over $n = 1, 2, 3, 4$ and $n = 5, 6, 7, 8$ respectively with $\{F_n^{(v)}(k_{\parallel}, z)\}$ being the associated envelope functions $\mathbf{k}_{\parallel} (= k_x \hat{\mathbf{x}} + k_y \hat{\mathbf{y}})$ is the in-plane wavevector with $\hat{\mathbf{x}}$ and $\hat{\mathbf{y}}$ unit vectors; A the normalisation area of the quantum well; $\{|v'\rangle\}$ the transformed zone centre Bloch functions while $\vec{\rho} = x\hat{\mathbf{x}} + y\hat{\mathbf{y}}$. The decoupled Hamiltonians are [16]

$$\mathbf{H}^{\pm}(k_{\parallel}, \hat{k}_z) = \begin{pmatrix} f + S_{hyd}^C & \mp g & \pm \frac{g}{\sqrt{3}} + h & \pm \sqrt{\frac{2}{3}}g + \frac{h^*}{\sqrt{2}} \\ \mp g & a_+ + S_{hyd}^V + S & c \pm iK & \sqrt{2}c \mp i \frac{K}{\sqrt{2}} \\ \pm \frac{g}{\sqrt{3}} + h^* & c \mp iK^{\dagger} & a_- + iJ + S_{hyd}^V - S & e \pm i\sqrt{\frac{3}{2}}L + \sqrt{2}S \\ \pm \sqrt{\frac{2}{3}}g + \frac{h}{\sqrt{2}} & \sqrt{2}c \pm i \frac{K}{\sqrt{2}} & e \mp i\sqrt{\frac{3}{2}}L^{\dagger} + \sqrt{2}S & d \mp iJ + S_{hyd}^V \end{pmatrix} \quad (2)$$

with

$$\begin{aligned} f &= E_{CO} + \left(\frac{\hbar^2}{2m_0}\right) \left[(2F+1)k_{\parallel}^2 + \hat{k}_z(2F+1)\hat{k}_z \right] \\ a_+ &= -E_{VO} - \left(\frac{\hbar^2}{2m_0}\right) \left[(\gamma_1 + \gamma_2)k_{\parallel}^2 + \hat{k}_z(\gamma_1 - 2\gamma_2)\hat{k}_z \right] \\ a_- &= -E_{VO} - \left(\frac{\hbar^2}{2m_0}\right) \left[(\gamma_1 - \gamma_2)k_{\parallel}^2 + \hat{k}_z(\gamma_1 + 2\gamma_2)\hat{k}_z \right] \\ d &= -E_{VO} - \Delta_O - \left(\frac{\hbar^2}{2m_0}\right) \left[\gamma_1 k_{\parallel}^2 + \hat{k}_z \gamma_1 \hat{k}_z \right] \\ b &= \left(\frac{\hbar^2}{2m_0}\right) 2\sqrt{3} k_{\parallel} \{ \gamma_3, \hat{k}_z \} \\ c &= \left(\frac{\hbar^2}{2m_0}\right) \sqrt{3} \gamma' k_{\parallel}^2; \quad \gamma' = \begin{cases} \frac{1}{2}(\gamma_2 + \gamma_3) & \text{axial approximation} \\ \gamma_2 & \text{along } \langle 100 \rangle \\ \gamma_3 & \text{along } \langle 110 \rangle \end{cases} \\ e &= \left(\frac{\hbar^2}{2m_0}\right) \sqrt{2} \left[2\hat{k}_z \gamma_2 \hat{k}_z - \gamma_2 k_{\parallel}^2 \right] \\ g &= \frac{1}{\sqrt{2}} P k_{\parallel} \\ h &= \sqrt{\frac{2}{3}} i \{ P, \hat{k}_z \} \\ J &= \left(\frac{\hbar^2}{2m_0}\right) k_{\parallel} \left[\hat{k}_z(\sigma - \delta - \pi) - (\sigma - \delta - \pi)\hat{k}_z \right] \\ K &= \left(\frac{\hbar^2}{2m_0}\right) \sqrt{3} k_{\parallel} \left[(\sigma - \delta)\hat{k}_z + \hat{k}_z(\pi) \right] \\ L &= \left(\frac{\hbar^2}{2m_0}\right) \sqrt{3} k_{\parallel} \left\{ \left[\frac{1}{3}(\sigma - \delta) + \frac{2}{3}\pi \right] \hat{k}_z + \hat{k}_z \left[\frac{2}{3}(\sigma - \delta) + \frac{1}{3}\pi \right] \right\} \end{aligned} \quad (3)$$

where

$$\begin{aligned} \delta &= \frac{1}{9} (1 + \gamma_1 + \gamma_2 - 3\gamma_3) \\ \sigma &= \frac{1}{2} (\gamma_3 + \gamma_2) - \frac{1}{2} \delta \\ \pi &= \frac{1}{2} (\gamma_3 - \gamma_2) + \frac{3}{2} \delta. \\ k_{\parallel}^2 &= k_x^2 + k_y^2; \quad \hat{k}_z = -i \frac{d}{dz} \end{aligned} \quad (4)$$

E_{CO} (E_{VO}) is the conduction (valence) band edge at the G-point, D_O the spin orbit splitting while P and F are, respectively, the first and second order Kane parameters for the conduction band. F is related to the experimentally determined conduction band effective mass m_c^* by [17]

$$\frac{1}{m_c^*} = (2F+1) + \frac{E_P}{3} \left[\frac{2}{E_g} + \frac{1}{E_g + \Delta_O} \right] \quad (5)$$

where E_g is the fundamental gap and $E_P = (2m_0 P^2) / \hbar^2$.

S_{hyd}^C and S_{hyd}^V are the strain induced hydrostatic shifts of the conduction and valence band edges respectively and S the shear strain shift. The parameters γ_1 , γ_2 and γ_3 are related to the Luttinger gamma parameters (γ_1^L , γ_2^L and γ_3^L) via the relations

$$\begin{aligned}\gamma_1^L &= \gamma_1 + E_p / (3E_g) \\ \gamma_2^L &= \gamma_2 + E_p / (6E_g) \\ \gamma_3^L &= \gamma_2 + E_p / (6E_g)\end{aligned}\quad (6)$$

The envelope functions satisfy

$$\sum_{v'=1}^4 \left[H_{vv'}^+ \left(k_{\parallel}, -i \frac{d}{dz} \right) \right] F_n^{(v')} (k_{\parallel}, z) = E_n(k_{\parallel}) F_n^{(v)} (k_{\parallel}, z) \quad (7)$$

and

$$\sum_{v'=1}^4 \left[H_{vv'}^- \left(k_{\parallel}, -i \frac{d}{dz} \right) \right] F_n^{(v'+4)} (k_{\parallel}, z) = E_n(k_{\parallel}) F_n^{(v+4)} (k_{\parallel}, z) \quad (8)$$

for $n = 1, 2, 3, 4$.

$H_{vv'}^{\pm}$ in Equation 8 are the components of the upper (+) and lower (-) blocks of the multi-band effective mass Hamiltonian [Eq.2] respectively. Equations (7) and (8) are solved by basis function expansion in terms of solutions, obtained by an accurate shooting method, of the corresponding single band problems. Details of the calculation are given in [16].

Optical Properties of Strained QW Lasers

Using the output from the bandstructure calculation, the optical gain at photon energy $\hbar\omega$ is calculated according to the expression [18]

$$g(\hbar\omega) = \frac{e^2 \pi}{n_w c \epsilon_o m_o^2 \omega L_z} \sum_{\sigma\sigma'} \sum_{nm} \int |\lambda \cdot M_{nm}^{\sigma\sigma'}(k_{\parallel})|^2 \times \frac{[f_n^C(k_{\parallel}) - f_m^V(k_{\parallel})](\hbar/\pi \tau_{in})}{(E_n^C(k_{\parallel}) - E_m^V(k_{\parallel}) - \hbar\omega)^2 + \left(\frac{\hbar}{\tau_{in}}\right)^2} \frac{k_{\parallel} dk_{\parallel}}{2\pi} \quad (9)$$

where n_w is the refractive index, m_o is the free electron mass, ϵ_o is the free space dielectric constant, c is the velocity of light, λ the polarisation vector of the optical field and e the magnitude of the electronic charge. $f_n^C(k_{\parallel})$ and $f_m^V(k_{\parallel})$ are the electron occupation probabilities for the conduction and valence bands respectively with τ_{in} the intra-band relaxation time, assumed here to be 10^{-13} s.

$\hbar = h/2\pi$ where h is Plank's constant. $E_n^C(k_{\parallel})$ and $E_m^V(k_{\parallel})$ are the conduction and valence subband energies respectively and $M_{nm}^{\sigma\sigma'}(k_{\parallel})$ the momentum matrix elements between conduction subbands n and valence subbands m . The momentum matrix elements are defined in [10].

The rate of radiative transitions due to the spontaneous emission process is [18]

$$R_{sp} = \int r_{sp}(\hbar\omega) d(\hbar\omega) \quad (10a)$$

where

$$r_{sp}(\hbar\omega) = \frac{e^2 n_w \omega}{\pi \hbar c^3 \epsilon_o m_o^2 L_z} \sum_{\sigma\sigma'} \sum_{nm} \int |M_{nm}^{\sigma\sigma'}(k_{\parallel})|_{sp}^2 \times \frac{f_n^C(k_{\parallel}) [1 - f_m^V(k_{\parallel})]}{(E_n^C(k_{\parallel}) - E_m^V(k_{\parallel}) - \hbar\omega)^2 + \left(\frac{\hbar}{\tau_{in}}\right)^2} \frac{k_{\parallel} dk_{\parallel}}{2\pi} \quad (10b)$$

with $M_{nm}^{\sigma\sigma'}(k_{\parallel})_{sp}$ being the spontaneous emission optical matrix elements [10]. The radiative component of the current density is then evaluated using

$$J^R = e L_z R_{sp} \quad (11)$$

The Auger component of the current density (J^A) is given by [19]

$$J^A = e L_z C (n^{3D})^3 \quad (12)$$

where C is the three-dimensional (3D) Auger coefficient, the 3D carrier density and L_z is the well width. In lasers, the threshold condition occurs when the peak gain equals the total losses in the optical cavity. The total current density is the sum of J^R and J^A .

3. NUMERICAL RESULTS AND DISCUSSION

This section presents results that demonstrate the improvement of threshold current performance in bipolar diode lasers through well shaping. Laser characteristics of tensilely strained InGaAs/InGaAsP (on InP) square and shaped quantum wells with an emission wavelength of 1.52mm are compared. At this wavelength, Auger recombination is the dominant carrier loss mechanism at room temperature, contributing up to 80% of the total threshold current [12,13,14]. In this work, only the influence of stepped alloy growth on laser performance is examined. Well shaping techniques such as delta-doping [20] and inter-diffusion [21] are not considered.

Energy Bandstructure

The energy bandstructure was calculated following the model outlined in the previous section. A 175Å tensilely strained $\text{In}_{0.36}\text{Ga}_{0.64}\text{As}$ square quantum well (SQW) with $\text{In}_{0.91}\text{Ga}_{0.09}\text{As}_{0.2}\text{P}_{0.8}$ barriers is taken as the reference system. This SQW has transition energy of 815 meV, equivalent to an emission wavelength of 1.52mm. The non-square quantum well considered here is designed to be much wider than the SQW yet within the critical layer thickness limit. The critical thickness is calculated using the expression proposed by Matthews and Blakeslee [22].

Figure 1 illustrates the conduction and light hole valence band profiles for the square and non-square quantum wells. The heavy-hole and split-off hole potential profiles are similar to the light-hole profile shown. The non-square quantum well is referred to as 'spiked quantum well' (SpQW) for obvious reasons. The heavy-hole (HH) and split-off-hole (SO) well profiles are similar to the light-hole profile. The SpQW has a wider well than the SQW, i.e. $L_2 > L_1$. Material compositions of the specific structures considered in this paper are given in Table 1.

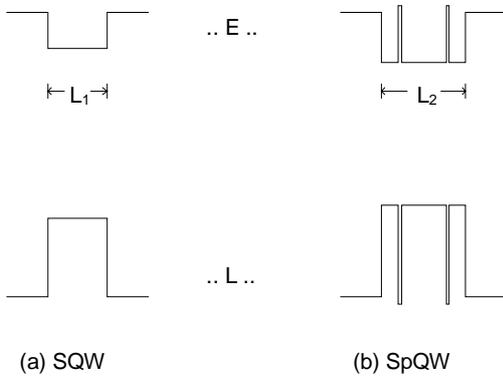


Figure 1: Schematic diagram of the electron (EL) and light-hole (LH) band edge profiles for (a) a square quantum well (SQW) and (b) a 'spiked' quantum well (SpQW).

Table 1. Material parameters for the spiked quantum well (SpQW) and the two square quantum wells (SQWs).

	A	B	D	E
$E_g (eV)$	0.936	0.888	1.274	1.201
$\Delta E_c^D (eV)$	-0.115	-0.139	—	—
$\Delta E_c^E (eV)$	-0.086	-0.112	—	—
m_c^*	0.049	0.047	0.076	0.073
γ_1^L	9.27	9.59	5.38	5.73
γ_2^L	3.45	3.6	1.76	1.89
γ_3^L	4.09	4.25	2.24	2.41
$S_{hyd}^c (meV)$	-81.4	-61.2	0	0
$S_{hyd}^v (meV)$	14.9	11	0	0
$S (meV)$	-45.3	-34.6	0	0
$\Delta_o (eV)$	0.322	0.32	0.149	0.167
$E_P (eV)$	24.3	24.3	24.3	24.3

The barrier material in the SpQW is unstrained 1.20 eV $\text{In}_{0.91}\text{Ga}_{0.09}\text{As}_{0.2}\text{P}_{0.8}$. The structure has two unstrained 1.27 eV, 12Å wide $\text{In}_{0.95}\text{Ga}_{0.05}\text{As}_{0.1}\text{P}_{0.9}$ spikes separated by 152Å incorporated symmetrically into the 254Å wide $\text{In}_{0.4}\text{Ga}_{0.6}\text{As}$ well region. The composition and separation of the spikes are chosen to keep the emission wavelength fixed at 815 meV while at the same time maintaining as large a separation as possible between the ground and first excited state electron, heavy-hole, light-hole and split-off hole subbands. The use of the potential spikes makes possible the design of quantum wells that are much wider than is possible with conventional square wells at a fixed emission wavelength, as determined by critical thickness constraints.

The room temperature material parameters for the two structures have been interpolated from the data of constituent binary alloys given in [23] and are shown in Table 1. Bowing parameters for the energy gaps are taken from [24] while the unstrained valence band offset for the InGaAs-InGaAsP interface is calculated following Weihsen [25] ΔE_c^D and ΔE_c^E are the strained conduction band offsets with respect to barrier materials D ($\text{In}_{0.95}\text{Ga}_{0.05}\text{As}_{0.1}\text{P}_{0.9}$) and E ($\text{In}_{0.91}\text{Ga}_{0.09}\text{As}_{0.2}\text{P}_{0.8}$) respectively.

Material A is $\text{In}_{0.36}\text{Ga}_{0.64}\text{As}$, B is $\text{In}_{0.4}\text{Ga}_{0.6}\text{As}$, and D is $\text{In}_{0.95}\text{Ga}_{0.05}\text{As}_{0.1}\text{P}_{0.9}$ while E is $\text{In}_{0.91}\text{Ga}_{0.09}\text{As}_{0.2}\text{P}_{0.8}$. The well regions of the 175Å and 150Å wide structures are formed from materials A and B respectively, while the 254Å SpQW has a material B well region and material D spikes. The barrier layers for all the structures are $\text{In}_{0.91}\text{Ga}_{0.09}\text{As}_{0.2}\text{P}_{0.8}$

(material E). The parameters are defined in the text.

The 175Å SQW is just below the calculated critical thickness and is theoretically the widest square quantum well attainable at 1.52 μm . Based on the trends reported in [1,11], namely that the threshold current falls with increasing well width and strain, it is expected that the 175Å SQW will have the lowest threshold current available in an SQW in this material system at 1.52 μm . A non-optimal, 150Å wide, 1.52 μm SQW of the same material composition ($\text{In}_{0.4}\text{Ga}_{0.6}\text{As}$) as the strained region of the SpQW has also been included in the study for purposes of comparison.

Figures 2 and 3 show the calculated valence band structure for the 254Å SpQW (dashed lines in each case) compared with the valence bandstructure of the 150Å SQW and the 175Å SQW (solid lines in each case). The zero of energy is taken to coincide with the zone centre energy of the first hole subband. From these figures one can see that for the SpQW, separation between the first two hole subbands is comparable to that of the two SQWs. Also, the in-plane mass (as reflected by the subband curvature) of the ground state subband in the SpQW is seen to be nearly equal to that in the 150Å SQW, but significantly heavier than that in the 175Å SQW. The gain and radiative current density characteristics based on the bandstructures of Figures 2 and 3 are presented next.

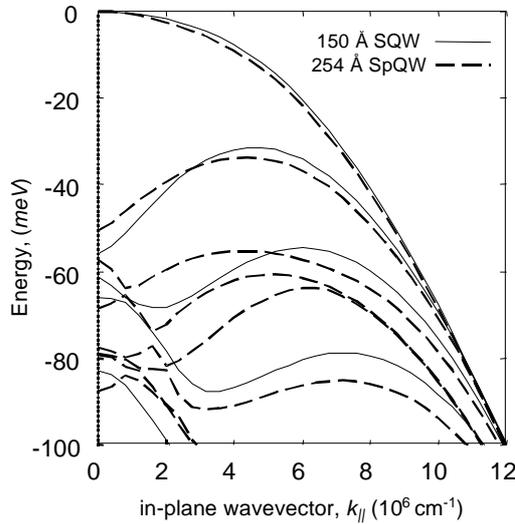


Figure 2: Variation of subband energy with in-plane wavevector (k_{\parallel}) for the 150Å SQW (solid lines) and the 254Å SpQW (dashed lines) described in Figure 1 and Table 2. The ordering of the three highest lying subbands is LH1-HH1-LH2 for the SQW and LH1-LH2-HH1 for the SpQW.

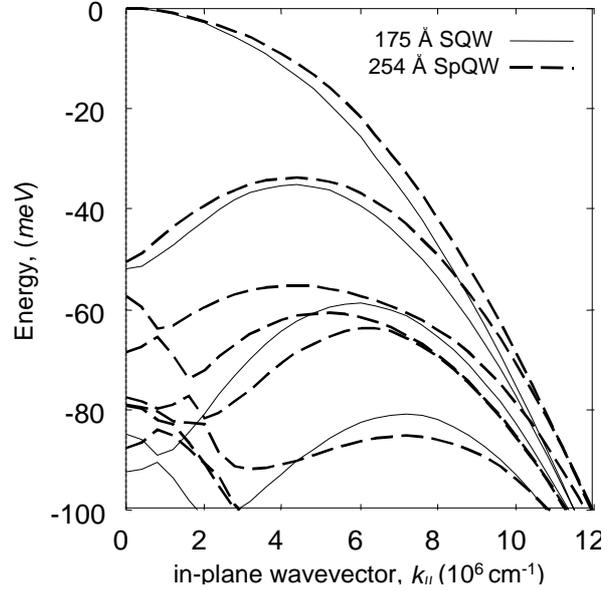


Figure 3: Variation of subband energy with in-plane wavevector (k_{\parallel}) for the 175Å SQW (solid lines) and the 254Å SpQW (dashed lines) described in Figure 1 and Table 2. The ordering of the three highest lying subbands is LH1-LH2-HH1 for the both SQW and the SpQW.

Optical Gain and Radiative Current Density Characteristics

The optical gain and radiative density characteristics were calculated using Equations 9 and 10 using energy dispersions and optical matrix elements from the bandstructure calculations. Since the quantum wells have different widths, the modal gain (Gg), rather than the material gain (g), is plotted to facilitate comparison. For a single step index separate confinement heterostructure (SCH) laser with a single quantum well of width L_z and an optical cavity of width d , the TM optical confinement factor (Γ_{TM}) is given by

$$\Gamma_{TM} = \left(\frac{n_b}{n_w} \right)^2 \frac{L_z}{d}$$

Where n_b is the barrier refractive index. Only the TM mode is considered since TE transitions are suppressed in tensilely strained structures [3]. The refractive indices have been calculated using the modified single effective oscillator model [11,12] and are listed in Table 2 along with the optical confinement factor calculated for a laser having an optical cavity width of 0.45mm. The barrier refractive index is 3.222 in all cases.

Table 2. Calculated refractive index and optical confinement factor for the 150Å and 175Å square quantum wells (SQWs) and the 254Å spiked quantum well (SpQW).

Structure	Refractive Index (n_w)	Optical Confinement Factor (Γ)
150Å SQW	3.481	0.0285
175Å SQW	3.466	0.0336
254Å SpQW	3.452	0.0491

Figure 4 shows the variation of modal TM gain with radiative current density (RCD) for the three structures. The SpQW has the highest radiative current density for all values of modal gain. At low modal gain, the 150Å SQW has marginally lower radiative current density performance than the 175Å SQW. This situation is reversed from about 15cm^{-1} , when the 175Å structure gives significantly lower RCD values with increasing modal gain. These results can be explained using the energy dispersions of Figures 2 and 3. Close spacing of the subbands leads to significant contribution of the excited states to the spontaneous emission and hence the radiative current. The 175Å SQW has the largest subband separations and therefore has the best characteristics overall. The close spacing of the excited state subbands of the SpQW is responsible for its comparatively poor performance.

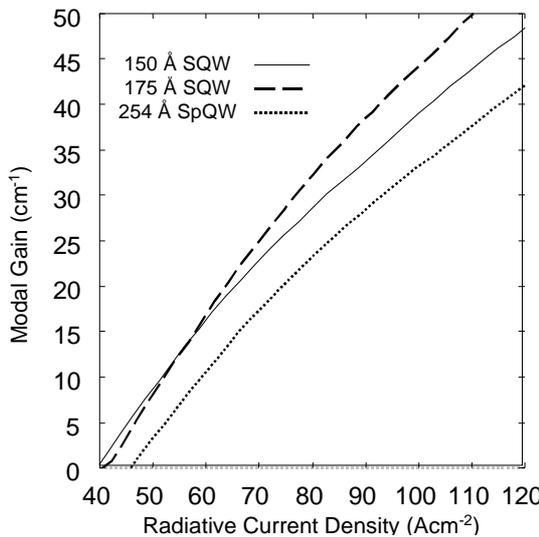


Figure 4: Modal TM gain as a function of radiative current density (RCD) for the 150Å and 175Å SQWs and the 254Å SpQW for a separate confinement heterostructure (SCH) laser with an optical cavity width of 0.45mm.

From these results, it is clear that the shaped quantum well structure presented in this paper is not expected to help reduce the radiative component of the threshold current compared to the optimised SQW. The radiative current component is however expected to be small (~20% of total current) compared to the Auger component [12,13,14]. It is now shown that the SpQW has lower Auger current density than the optimal SQW in low loss laser cavities, leading to lower total threshold current density.

In their study of tensilely strained InGaAs-InGaAsP laser structures, Jones *et al.*[28] show that quantum well structures having well widths greater than ~150Å exhibit bulk-like behaviour. Since the three structures in this study satisfy this criterion, the calculated gain has been plotted against the three-dimensional carrier density as opposed to the usual two-dimensional carrier density. The variation of modal TM gain with the 3D carrier density is shown in Figure 5 for the 150Å SQW (solid line), the 175Å SQW (dashed line) and the 254Å SpQW (dotted line). It is seen that the 254Å SpQW has the highest values of modal gain at any given value of 3D carrier density compared with the two SQWs.

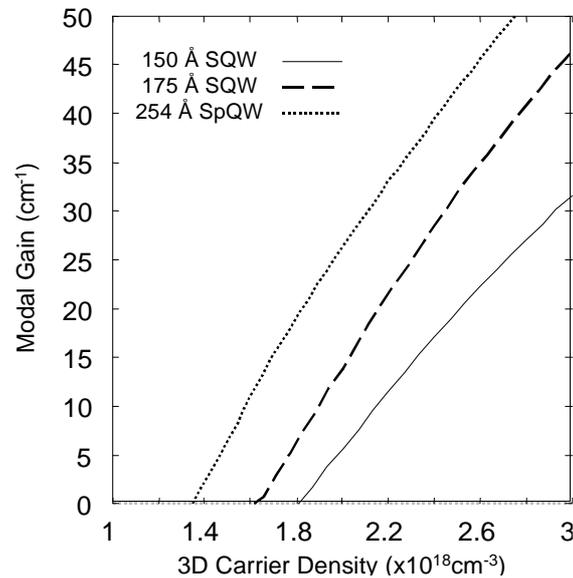


Figure 5: Plots of the modal TM gain as a function of the 3D carrier density for the 150Å and 175Å SQWs and the 254Å SpQW for a separate confinement heterostructure (SCH) laser with an optical cavity width of 0.45mm.

While the radiative current density (RCD) at threshold can be determined directly from the gain/RCD characteristics (Figure 4), evaluation of the dominant Auger current density requires knowledge of the Auger coefficient (C). The Auger current is calculated using Equation 12. Values

of C vary from one material system to another and depend on the type of recombination process which is dominant, i.e., whether direct or phonon assisted. A number of authors have published values of Auger coefficients for InGaAs and InGaAsP quantum well structures with 1.55 μ m emission wavelength [29, 30, 31, 32]. However, most of the work reported in these references focuses on quantum well structures under compressive strain. Relatively little systematic work has been done for tensilely strained structures [12]. There clearly is a need to characterise experimentally the Auger coefficients of quantum well structures for a wider range of material compositions and systems. Given that the structures studied in this work are under tensile strain, and in the light of the shortcomings just presented, only a comparative assessment of their performance as lasers is presented.

Evaluation

It has been shown both theoretically and experimentally that the value of Auger coefficient (C), at worst, remains constant as the well width is increased while keeping the emission wavelength constant [29,30,31]. In the calculations that follow therefore, a value of C is assumed for the two SQWs and the SpQW. The Auger component of the threshold current is calculated using the data of Figure 5 and Equation 12 for the different values of modal gain. The total current is obtained by adding the Auger component to the radiative component of Figure 4. The result is shown in Figure 6.

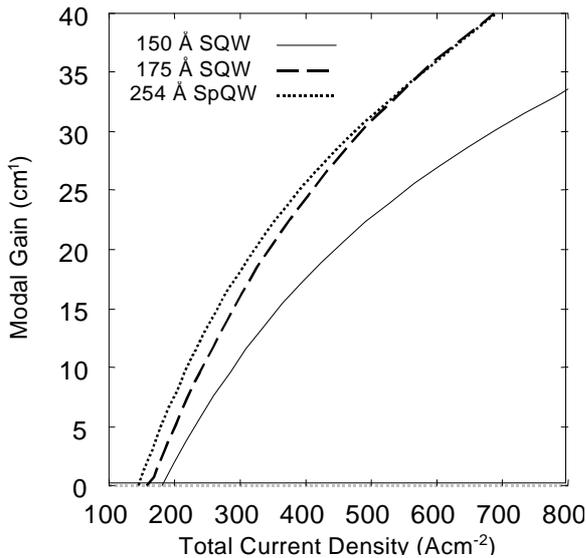


Figure 6: Variation of modal TM gains with total current density for the 150Å and 175Å SQWs and the 254Å SpQW for a separate confinement heterostructure (SCH) laser with an optical cavity width of 0.45mm.

Of the three structures, the 150Å SQW has the highest values of total current density for all values of modal gain. The 254Å SpQW has the lowest values of total current density for lasers with cavity losses between 0 cm^{-1} and $\sim 30\text{cm}^{-1}$ while, beyond $\sim 30\text{cm}^{-1}$, the 175Å has the lowest total current density. A laser incorporating an SpQW will therefore have the best threshold current performance at 1.52 μ m in the low loss regime. Table 3 shows the calculated total gain for the three structures at modal gain values of 5 cm^{-1} , 10 cm^{-1} and 20 cm^{-1} .

Table 3: Calculated values of total threshold current density at modal threshold gain values of 5 cm^{-1} , 10 cm^{-1} and 20 cm^{-1} for the two SQWs and for the SpQW.

	TOTAL THRESHOLD CURRENT DENSITY (Acm^{-2})		
	$g_{th}^m = 5\text{cm}^{-1}$	$g_{th}^m = 10\text{cm}^{-1}$	$g_{th}^m = 20\text{cm}^{-1}$
150Å SQW	238	285	457
175Å SQW	202	240	351
254Å SpQW	180	218	321

Compared to the 175Å SQW, the 254Å SpQW design gives reductions in total current density of 10.9% (at 5 cm^{-1}), 9.2% (at 10 cm^{-1}) and 8.5% (at 20 cm^{-1}). It is worth noting that the results shown in Figure 6 and Table 3 represent the worst-case scenario for the wide, 254Å, SpQW as its Auger coefficient C is expected to be smaller than that of the narrower SQWs. Using a smaller value of C for the SpQW should result in a larger reduction of the total current density for the SpQW.

4. SUMMARY AND CONCLUSION

The electronic and optical properties of quantum wells of arbitrary shape have been calculated using a multi-band effective mass model that explicitly incorporates coupling between the lowest lying conduction band and the highest lying valence bands. It has been shown that well shaping by stepped alloy growth can be beneficial for reducing the threshold current in bipolar diode lasers operating at a wavelength of 1.52 μ m.

At 1.52 μ m, Auger current (J^A) dominates the laser threshold current and since $J^A \sim L_z C(n^{3D})^3$, a reduction in the 3D carrier density n^{3D} will result in lower threshold current laser operation. For tensilely strained InGaAs-InGaAsP structures, shaping of the quantum wells using a potential 'spikes' produces a significantly wider well than an optimal square quantum well of the same emission

wavelength and barrier composition. This leads to a reduced 3D carrier density, and so a reduction of the Auger current for low loss bipolar diode lasers. This prediction is arrived at by assuming that the 3D Auger coefficient is independent of well width for relatively wide wells [29,30]. The specific 254Å spiked quantum well (SpQW) considered has a smaller value of 3D carrier density at all values of modal gain than an optimal 175Å square quantum well (SQW). On the other hand, the modal gain/radiative current density characteristics are not improved for the SpQW, leading to a larger value of radiative threshold current density than for the SQW laser. However, since Auger current dominates, the increased radiative current component of the SpQW is more than offset by the reduced Auger component, leading to reduced total threshold current density for a low loss, bipolar diode laser.

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